Study on structural recovery of graphite irradiated with swift heavy ions at high temperature


A R T I C L E   I N F O

Article history:
Received 29 May 2015
Received in revised form 28 August 2015
Accepted 1 September 2015
Available online 12 September 2015

Keywords:
High-power graphite target
Swift heavy ion
Radiation damage
Structural recovery

A B S T R A C T
Thin graphite foils bombarded with an intense high-energy (8.6 MeV/u) gold beam reaching fluences up to $1 \times 10^{15}$ ions/cm$^2$ lead to swelling and electrical resistivity changes. As shown earlier, these effects are diminished with increasing irradiation temperature. The work reported here extends the investigation of beam induced changes of these samples by structural analysis using synchrotron X-ray diffraction and transmission electron microscope. A nearly complete recovery from swelling at irradiation temperatures above about 1500 °C is identified.

1. Introduction

The realization of a production target for the new generation of high-energy high-intensity heavy ion beam facilities such as the rare isotope production at the Facility for Rare Isotope Beams (FRIB) [1] in the USA and at the Facility for Antiproton and Ion Research (FAIR) [2] in Germany is challenging. The target material has to withstand the combined effects of high power density beams generating thermo-mechanical stress as well as radiation damage leading to structural modifications and changes of thermo-mechanical properties. All these effects will limit the reliable function and the lifetime of the targets.

In collaboration within the FRIB and FAIR project, thin graphite foils were irradiated at the GSI UNILAC accelerator with an intense 8.6 MeV/u gold beam at fluences up to $1 \times 10^{15}$ ions/cm$^2$. The samples showed swelling and an increase of the electrical resistivity. These beam-induced changes were observed to diminish with increasing irradiation temperature [3].

This paper presents an extended investigation of part of these irradiated graphite samples, using synchrotron X-ray diffraction (XRD) and High Resolution Transmission Electron Microscope (HRTEM). Measurements of the $d$ spacing provide new insights into graphite performance under extreme ion beam exposure. Applying a 8.6 MeV/u Au beam provides an electronic energy loss value of ~20 keV/nm [4], which is higher than the future FRIB beams (energy loss between 0.04 and 6 keV/nm). The target material is thus tested under a harsher energy density deposition than in future applied conditions.

2. Experimental procedure

The experimental set-up is defined in more detailed description in Ref. [3]. As samples we used thin foils of purified polycrystalline isotropic graphite (MERSEN grade 2320) with a nominal thickness of 75 μm. During the irradiation with 8.6 MeV/u Au ions, the electric current for ohmic heating was adjusted to the desired irradiation temperature [3].

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The samples were characterized before and after irradiation by synchrotron X-ray Diffraction measurements (XRD) performed at the Advanced Photon Source (Lawrence Berkeley Natl. Laboratory),...
using high energy X-ray (30 keV). The X-ray wavelength was 0.4132 Å.

High Resolution Transmission Electron Microscope (HRTEM) analysis was performed with an JEOL 3011 electron microscope at the University of Michigan. For TEM inspection, the irradiated samples were crushed and deposited on a carbon film supported by a copper TEM grid. Gold particles were intentionally deposited on top of the ground graphite specimens in order to calibrate the scale bars. During the TEM observations, the electron current density was kept as low as possible (0.1–1 A/cm²) to minimize electron-beam-induced microstructural changes.

3. Results and discussion

3.1. X-ray diffraction analysis

The diffraction patterns were recorded for graphite samples irradiated at different fluences and temperatures are shown in Fig. 1. Comparing the diffraction pattern of the pristine and the sample irradiated at 385 °C reveals a pronounced broadening of the (002) diffraction maximum. Peak broadening is in general attributed to a reduction of grain size [5] and/or build-up of micro-strain in the sample [6]. Moreover, the broadening of the (002) peak is accompanied by a shift to a smaller angle (Table 1), indicative for an increase of the d-spacing. In contrast, the (110) diffraction maximum is insensitive to the ion irradiation showing within our experimental accuracy neither peak broadening and nor a shifts (Table 1) which indicates no significant change of the in-plane lattice parameter.

With increasing irradiation temperature, the (002) diffraction maximum shifts back towards its original two theta value (pristine sample) concomitant with a reduction in width, revealing in situ defect annealing. The increase of the (002) peak intensity at higher temperatures and fluences. The bottom pattern is the pristine sample for reference and the top pattern is the sample irradiated to the highest fluence (1 x 10¹⁵ ions/cm²); the three patterns in-between are data from samples irradiated to the same fluence (1 x 10¹⁴ ions/cm²) but at different temperatures.

3.2. HRTEM analysis

A HRTEM image of a graphite sample irradiated at 1525 °C (fluence 1 x 10¹⁴ ions/cm²) together with the corresponding Fast Fourier Transform (FFT) analysis is presented in Fig. 2. Variations in lattice parameter are subtle for samples irradiated at different temperatures. Therefore, instead of directly measuring the d-spacing in the HRTEM images, gold particles with the known d-spacing were used to calibrate precisely the d-spacing of graphite. From the FFT image, the d-spacing of graphite (111) was determined by the constant R’d relation where R is the radius of a ring in the diffraction pattern. In a FFT image, a vector pointing from (0000) to a diffraction spot represents the corresponding crystal direction in the imaging plane. The lattice spacing of (111) gold is dAu111 = 2.35 Å, the lattice spacing of (111) carbon is derived by

\[ d_{C111} = d_{Au111} \times \left( \frac{R_{Au111}}{R_{C111}} \right) \]

The deduced values of the d-spacing of graphite are reported in Table 1 for each sample.

From HRTEM and synchrotron XRD analyses, we inferred the lattice parameter changes from d-spacing values of peak (002) and peak (111) which is relative to the changes in c-axis, with

![Synchrotron XRD patterns of graphite samples irradiated at different temperatures and fluences.](image)

**Fig. 1.** Synchrotron XRD patterns of graphite samples irradiated at different temperatures and fluences. The bottom pattern is the pristine sample for reference and the top pattern is the sample irradiated to the highest fluence (1 x 10¹⁵ ions/cm²); the three patterns in-between are data from samples irradiated to the same fluence (1 x 10¹⁴ ions/cm²) but at different temperatures.

Table 1

<table>
<thead>
<tr>
<th>Samples</th>
<th>Pristine</th>
<th>1 A</th>
<th>19 A</th>
<th>35 A</th>
<th>36 A</th>
</tr>
</thead>
<tbody>
<tr>
<td>Mean flux (ions/cm² s)</td>
<td>3.8 x 10¹⁰</td>
<td>4.5 x 10¹⁰</td>
<td>4.2 x 10¹⁰</td>
<td>5.6 x 10¹⁰</td>
<td></td>
</tr>
<tr>
<td>Irradiation temperature (°C)</td>
<td>385 ± 40</td>
<td>910 ± 150</td>
<td>1525 ± 95</td>
<td>1535 ± 275</td>
<td></td>
</tr>
<tr>
<td>d-Spacing (002) (Å)</td>
<td>3.4197</td>
<td>3.5415</td>
<td>3.5330</td>
<td>3.5466</td>
<td></td>
</tr>
<tr>
<td>Relative expansion of d-spacing (002) (%)</td>
<td>1.37 ± 0.50</td>
<td>1.32 ± 0.47</td>
<td>1.37 ± 0.59</td>
<td>1.30 ± 0.19</td>
<td></td>
</tr>
<tr>
<td>d-Spacing (110) (Å)</td>
<td>1.2438</td>
<td>1.2443</td>
<td>1.2426</td>
<td>1.2471</td>
<td></td>
</tr>
<tr>
<td>Relative expansion of d-spacing (110) (%)</td>
<td>0.04 ± 0.01</td>
<td>−0.10 ± 0.01</td>
<td>0.26 ± 0.15</td>
<td>−0.09 ± 0.01</td>
<td></td>
</tr>
<tr>
<td>d-Spacing (111) (Å)</td>
<td>3.47</td>
<td>3.38</td>
<td>3.30</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Relative expansion of d-spacing (111) (%)</td>
<td>3.17 ± 0.32</td>
<td>0.86 ± 0.09</td>
<td></td>
<td></td>
<td></td>
</tr>
</tbody>
</table>

The sample name denotes the heating current (in Amp) applied during irradiation. d-Spacing extracted from the Synchrotron XRD pattern (peak (002) and peak (111)) performed at Advanced Photon Source (Lawrence Berkeley Natl. Laboratory). d-Spacing extracted from HRTEM investigation (peak (111)) performed at University of Michigan. From these values, the relative expansion is inferred and compared to the pristine sample.
respect to the pristine sample as a function of irradiation temperature. Fig. 3 shows the relative lattice parameter change as a function of irradiation temperature for two fluences. Both XRD and HRTEM analysis revealed a consistent lattice expansion of 3–4% for an irradiation temperature up to about 1000 °C. Further increase of the sample temperature during irradiation results in less radiation damage. This in-situ annealing effect is evident in the XRD and HRTEM data and seems to act at both fluences tested. Extrapolating this result, we expect that irradiations performed at or above a temperature of 1900 °C avoid beam-induced structural changes of graphite.

Defect annealing is also supported by results from post-irradiation investigations of electrical resistance [3], thermal diffusivity [7] and Young's modulus [8] properties of heavy-ion irradiated graphite that had similar pristine properties and irradiation conditions as the ones used in this study. For the irradiation of similar polycrystalline fine-grain graphite with fast neutrons, it was also observed that defect annealing associated with partial property recovery occurs at irradiation temperatures above 1000 °C [9].

4. Conclusion

Gold ions of 8.6 MeV/u kinetic energy and high electronic energy loss (20 keV/nm) were used to irradiate polycrystalline graphite (MERSEN Grade 2320) at different fluences and irradiation temperatures. HRTEM and XRD analyses were performed and consistently identify an increase of the lattice parameter at irradiation temperatures up to about 1000 °C. For irradiations at 1535 °C, the lattice parameter expansion is less pronounced indicating defect annealing effects during irradiation. Target operation at such high temperature obviously mitigates radiation damage and make this type of graphite a suitable material for use in a high-power production target designed for rare isotope beam production at FRIB and FAIR.

Acknowledgments

The authors gratefully acknowledge technical support during synchrotron XRD at the Advanced Photon Source of Lawrence Berkeley National Laboratory. The Advanced Light Source is supported by the Director, Office of Science, Office of Basic Energy Sciences, of the U.S. Department of Energy under Contract No. DE-AC02-05CH11231. The TEM and XRD analysis was completed at the University of Michigan and supported by the Energy Frontier Research Center Materials Science of Actinides funded by the U.S. Department of Energy, Office of Science, Office of Basic Energy Sciences (DE-SC0001089).

This material is based upon work supported by the U.S. Department of Energy Office of Science under Cooperative Agreement DE-SC0000661, the State of Michigan and Michigan State University. Michigan State University designs and establishes FRIB as a DOE Office of Science National User Facility in support of the mission of the Office of Nuclear Physics.

References